

Figure 3.-Stereoscopic packing diagram showing the contents of two unit cells (stacked along the *c* direction) viewed down the *a** reciprocal axis with the *c* axis horizontal.

membered chelate ring with the nickel atom, is puckered. The carbon atom C(9) lies 0.40 *k* on one side of the Ni-O-N coordination plane while $C(8)$ is only 0.11 Å from this plane on the opposite side, as might be expected considering the idealized hybridizations of the oxygen $(sp³)$ and nitrogen $(sp²)$ donor atoms.

The benzo rings are unremarkable and are quite accurately planar with all atoms within 0.03 *k* of the least-squares plane. The C-C bonds of the benzo group average 1.39 A but exhibit sizable deviations from this mean value as has been found in other salicylaldimine chelates." The present bond distances may suffer to some extent from greater inaccuracies

owing to the presence of the heavy iodine atom which dominates the X-ray scattering.

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The Crystal Structure of BazMnSes. Linear Antiferromagnetism in $Ba₂MnX₃$ (X = S, Se)¹

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The compounds Ba₂MnSe₃ and Ba₂MnS₃ are isostructural with K₂AgI₃; the unit cell parameters for the selenide are $a =$ 9.135 (2) \hat{A} , $b = 4.471$ (1) \hat{A} , and $c = 17.736$ (2) \hat{A} and for the sulfide $a = 8.814$ (5) \hat{A} , $b = 4.302$ (2) \hat{A} , and $c = 17.048$ (8) Å measured at $25 \pm 0.5^{\circ}$, space group *Pnma, Z* = 4. Three-dimensional data were collected with Mo K α radiation, using balanced filters and the stationary-crystal, stationary-counter method to obtain peak heights. The structures were refined by least squares using anisotropic temperature factors to a final *R* value of 0.058 for 992 reflections for the selenide and to a final R value of 0.059 for 233 reflections for the sulfide using isotropic temperature factor parameters. The structure consists of Mn-X₄ tetrahedra sharing corners to produce infinite chains parallel to *b*. The magnetic susceptibility for both compounds was determined over the temperature interval 50-600°K by the Faraday technique. The Heisenberg model for a linear chain antiferromagnet and a "reduced-spin'' model were used to calculate the exchange interaction parameter J/k and the Landé g factor. The results from both models are in close agreement with $J/k = -12.3$ (5)^oK and g = 2.00 (3) for the sulfide and $-9.8(5)$ ^oK and $g = 2.03(3)$ for the selenide on the basis of the Heisenberg model.

Introduction

In the course of a study of the system BaX-M-X $(X = S, Se; M = Mn, Fe, Co, Ni)$ we recently at-**(1) Research sponsored by the Robert A. Welch Foundation, Houston,**

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tempted to prepare $BaMnSe_2$ analogously to $BaNiS_2$ and $BaCoS₂$ ³ We were not able to prepare a single-phase material. The reaction mixture was found to consist of crystals of manganese selenide (α -MnSe) mixed with

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Figure 1.-Susceptibility data for Ba₂MnS₃ (A) and Ba₂- $MnSe₃$ (B). The three sets of experimental points are results from three different preparations of A and B. The theoretical curve for A was calculated from eq 1 using $J/k = -12.3$ °K, $g = 2.00$ and for B using $J/k = -9.8$ °K, $g = 2.03$.

tabular orange crystals of an unknown phase,

A three-dimensional X-ray structural analysis showed that the composition was $Ba₂MnSe₃$. The compound has the K_2AgI_3 structure⁴ so that manganese selenide tetrahedra share corners to produce infinite chains [MnSe₂Se_{2/₂]_n⁴ⁿ⁻ parallel to the *b* axis. Neighboring} chains are well separated and so the compound is a good example for studying linear antiferromagnetism in a chain composed of magnetic ions with five unpaired electrons. Divalent manganese, with a large spin, $S = \frac{5}{2}$, is particularly well suited to an application of the Heisenberg model linear chain antiferromagnet in the classical limit $(S \rightarrow \infty)$ as developed by Fisher.⁵ In the classical limit of infinite spin on the interacting ions, the quantum mechanical operators in the Heisenberg model may be replaced by classical vectors, allowing an exact solution. The model has been modified expression

to apply to a Mn²⁺ system⁶ resulting in a susceptibility
expression

$$
\frac{1}{\chi_{m'}} = \frac{0.937}{g^2} T \left[\frac{1 + \coth 17.5x - 0.0571/x}{1 - \coth 17.5x + 0.0571/x} \right] (1)
$$

where *g* is the Landé factor, $x = -J/kT$, and J/k (in degrees Kelvin) is defined as negative for antiferromagnetic coupling.

Equation 1 has been used with some success in fitting susceptibility data for $[(CH_3)_4N]MnCl_3^7$ and CsMn- $Cl_3 \tcdot 2H_2O$.⁸ Theoretical analyses revealed that interchain interactions are many orders of magnitude smaller than intrachain exchange. For the latter compound corrections were made to account for the fact that the spin of manganese, although large, is not infinite as is formally required by the Fisher model. This correction was relatively minor, resulting in a change in the value of *J* of approximately 3% .

The compounds $Ba₂MnX₃$ (X = S, Se) are quite similar to $\text{CsMnCl}_3 \cdot 2\text{H}_2\text{O}$, with regard to the geometry

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of the Mn-X-Mn superexchange path and the Mn-Mn interchain and intrachain separations. We thus prepared pure samples of Ba₂MnSe₃ and the isomorphous $Ba₂ MnS₃$ and carried out magnetic susceptibility measurements in the temperature range $50-600^{\circ}$ K, to see how well the Fisher model applies to these compounds and to determine the effect of varying the covalency at the bridging ligand.

Experimental Section

by heating stochiometric mixtures BaSe: Mn:Se in sealed Vycor

tubes at temperatures ranging from 500 to 950°. In each case,

the product consisted of a mixture of black crystals (MnSe) to-

r. **'K**
 T. 'K (a) Preparation.---Attempts to prepare $BaMnSe_2$ were made by heating stoichiometric mixtures Base: Mn 'Se in sealed Vycor the product consisted of a mixture of black crystals (MnSe) together with orange tabular crystals, identified subsequently by single-crystal X-ray analysis as $Ba₂MnSe₃$.

Pure samples of Ba₂MnSe₃ and Ba₂MnS₃ were then prepared by allowing the appropriate stoichiometric mixtures to react at 850 and 1050°, respectively. The reaction mixtures were contained within carbon tubes to prevent reaction with the Vycor ampoule and after heating for 2-5 days the samples were slowly cooled to room temperature. The resulting products were highly crystalline and the crystals formed deep orange-red columns, The formation of single-phase material was confirmed by microscopic examination and X-ray powder diffraction methods.

(b) Magnetic Measurements.-The magnetic susceptibility of finely powered samples of the compounds was measured in the temperature range 50 (pumped solid nitrogen) to 600° K by the Faraday technique. The apparatus employed a 6-in. electromagnet fitted with 4-in. Faraday pole caps, together with a Cahn RG electrobalance, sensitive to 0.1 μ g. Separate, evacuable sample chambers were used for high- and low-temperature measurements. The magnetic field strength was determined individually for the separate chambers in place, using $\text{CoHg}(\text{SCN})_4$. The susceptibilities of the compounds were checked at a number of values of the field gradient ($H dH/dz$) between 1.4 \times 10⁶ and 10.5×10^6 kG² cm⁻¹. Temperature runs were made on three separately prepared samples of each of the compounds. The values of the susceptibilities from different runs were in agreement to $2-4\%$.

The experimental results are summarized in the plots of inverse susceptibility *vs.* temperature in Figure 1.

X-Rap Structure Determination

Zero and upper level Weissenberg photographs of single crystals selected from the "Ba $MnSe₂$ " reaction mixture showed diffraction symmetry *mmm* and the systematic absences were *Okl,* $k + l = 2n + 1$ and *hkO, h =* $2n + 1$ *, consistent with the space groups Pnma* and *PnaZ1.* **A** columnar crystal measuring $0.06 \times 0.05 \times 0.15$ mm was selected and mounted along its long dimension *(b* axis) on a GE single-crystal orienter. Eighty reflections were carefully centered on the K_{α_1} and K_{α_2} components of molybdenum radiation $(\lambda_1$ 0.70926 Å, λ_2 0.71354 Å) using a 1° takeoff angle and a 0.02° slit. These 2θ measurements were used in a least-squares refinement for the determination of the lattice parameters $a = 9.135$ (2) Å, $b = 4.471$ (1) Å, $c = 17.736$ (2) Å, at $25 \pm 0.5^{\circ}$.

Three-dimensional data to $(\sin \theta)/\lambda = 0.703$ were collected with Mo K_{α} radiation, a 5° takeoff angle, balanced filters, and the stationary-crystal, stationarycounter method. Peak heights were counted for 10 sec using a zirconium filter; then background was counted for 10 sec using an yttrium filter. A total of 1180 independent reflections were measured of which

TABLE I

^{*a*} The temperature factor is $\exp[-(\beta_{11}h^2 + \beta_{22}k^2 + \beta_{33}l^2 + 2\beta_{12}hk + 2\beta_{13}hl + 2\beta_{23}kl)].$

993 were considered observed. Lorentz and polarization corrections were applied and an absorption correction made (μ 295.1 cm⁻¹). The absorption factor ranged from 4.4 to 7.0 . The equation

$$
\sigma(F) = \frac{1}{2} \left\{ K \frac{[1 + I_{\rm Y}/I_{\rm Zr}]}{[1 - I_{\rm Y}/I_{\rm Zr}]} \right\}^{1/2}
$$

where I_Y and I_{Zr} are the number of counts with the Y and Zr filters and K is the product of the Lorentz, polarization, absorption, α splitting, and tube current corrections, was used to estimate the standard deviation of the observed structure factor.

The choice of the centrosymmetric space group, *Pnma*, was made on the basis of the statistical distribution of the structure factors obtained from a Wilson plot calculation which also supplied approximate scale and temperature factors. The symbolic addition method for the direct determination of phases was used in the solution of the structure. The sequence of computer programs FAME-MAGIC-LINK-SYMPL⁹ was used to generate 136 phases from which an E map was calculated showing clearly three selenium and two barium atoms. The E map was complicated by a pseudomirror plane at $x = 0.25$ but the correct set of atom positions was determined from packing considerations.

From a structure factor calculation based on the barium and selenium positions a discrepancy value, R , of 0.22 was obtained. A Fourier map revealed the position of the manganese atom. The atom parameters, scale, and isotropic temperature factors were then refined using a least-squares program XFLS, a modification of ORFLS.¹⁰ The refinement was based on F_0 and the structure factors were weighted by $1/\sigma^2$. After four cycles of refinement the value of R was 0.082. At this stage the isotropic temperature factors were converted to anisotropic factors and three more cycles of refinement led to a final R value of 0.070 for all measured reflections, with $R = 0.058$ and $wR = 0.069$ for nonzero reflections only. Atomic scattering factors

were taken from ref 11 and were corrected for $\Delta f'$ and $\Delta f''$. The standard deviation of an observation of unit weight was 1.55. A three-dimensional difference map was calculated and showed no peaks greater than $0.5 e^-/\AA^3$.

The final positional parameters and anisotropic temperature factors are given in Table I. The bond distances and angles are given in Table II and Table III lists the F_0 and F_0 values.

TABLE II BOND DISTANCES AND ANGLES IN Ba2MnSe3

	Distances, A		
$Mn-Mn(intrachain)$	4.471(1)	$Ba_1 - Se_1(2)$	3.295(4)
$Mn-Mn(interchain)$	6.164(6)	$Ba_1 - Se_1(2)$	3.282(4)
		$Ba_1 - Se_2(2)$	3.382(4)
$Mn-Se_1(1)$	2.538(6)	$Ba_1-Se_8(1)$	3.399(4)
$Mn-Se_2(1)$	2.524(6)		
$Mn-Se3(2)$	2.601(6)	$Ba_2-Se_1(1)$	3.304(4)
		$Ba_2-Se_2(1)$	3.376(4)
$Se1-Se2(1)$	3.978(5)	$Ba_2-Se_2(2)$	3.211(4)
$Se_3-Se_3(1)$	4.471	$Ba2-Se3(1)$	3.473(4)
$Se1-Se3(2)$	4.177(5)	$Ba_2-Se_3(2)$	3.382(4)
$Se_2-Se_3(2)$	4.150(5)		
	Angles, Deg		
$Se_1-Mn-Se_2(1)$	103.6(2)	$Se1-Mn-Se3(2)$	108.7(2)
$Se_3-Mn-Se_3(1)$	118.5 (2)	$Se2-Mn-Se3(2)$	108.1(2)

X-Ray powder diffractograms for the corresponding sulfide, Ba₂MnS₃, were closely similar to those of Ba₂-MnSe₃ and indicated that they are isostructural. Unit cell parameters were determined from a least-squares fit to the 2θ values for 15 nonoverlapping powder diffraction lines, obtained on a Philips powder diffractometer, using Cu K α radiation, λ 1.542 Å, and are α = 8.814 (5), $b = 4.302$ (2), and $c = 17.048$ (8) Å, at 25 \pm 0.5°. A single crystal of Ba₂MnS₃, whose largest dimension was approximately 50 μ , was mounted and three-dimensional reflection data to $2\theta = 40^{\circ}$ were collected with Mo K_{α} radiation using the same experimental conditions as for Ba₂MnSe₃ and the reduction of the data was also the same. The coordinates ob-

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tained for the selenide were used as a basis for leastsquares refinement with isotropic temperature factors, and a final discrepancy value, $R = 0.058$, for all reflections (233) was obtained. The atomic parameters are listed in Table I and bond distances and angles are given in Table IV; Table V lists F_0 and $|F_0|$.

TABLE IV BOND DISTANCES AND ANGLES IN Ba₂MnS₈

	DOND DISTANCES AND INNOCES IN DAZIMIOS				
\leftarrow \leftarrow \leftarrow \leftarrow \leftarrow \leftarrow					
$S_1-Mn-S_2(1)$	103.3(9)				
$S_3 - Mn - S_3(1)$	119.8(9)				
$S_1-Mn-S_3(2)$	108.1(9)				
$S_2-Mn-S_3(2)$	108.1(9)				
	5.96(2) 2.42(3) 2.39(3) 2.48(3) 3.78(3) 3.97(3) 3.95(3)				

Discussion

The structure of Ba₂MnSe₃ is illustrated in the stereographic drawing in Figure 2. It consists of infinite linear chains of $[MnSe_2Se_{1/2}]_n^{4n-}$ corner-sharing tetrahedra, parallel to the *b* axis, with barium ions packing between the chains. The two crystallographically independent barium ions have seven selenium atoms as nearest neighbors although the shapes of the coordination polyhedra differ markedly. The $Ba₁$ ion is at the

Figure 2.-Stereoscopic drawing of $Ba₂MnSe₃$. The unit cell outline is shown; the vertical axis is *b* and the horizontal axis is c. The structure is shown after a 15° rotation about c.

center of a trigonal prism of selenium ions with the seventh selenium ion projecting from the center of one of the rectangular faces of the prism. All seven selenium ions belong to two neighboring linear chains. The Ba₁-Se distances range from 3.282 (4) to 3.399 (4) \AA . The second barium ion coordinates to selenium ions from four surrounding linear chains at distances ranging from 3.211 (4) to 3.473 (4) **8.** Within a linear chain, the manganese ions are at the centers of distorted tetrahedra of selenium ions. The bonds to the selenium ions involved in the corner sharing, Mn-Se₃, of length 2.601 (6) **8,** are slightly longer than the other two Mn-Se bonds, 2.524 (6) and 2.538 (6) *k.* The latter two distances are equal within the accuracy of the experiment and they subtend almost equal angles with Mn-Se₃, which are close to the tetrahedral angle, 109.5° . However there is a considerable distortion of the bridging angle, $Se_3-Mn-Se_3$, and the angle involving the nonbridging seleniums, $Se₁-Mn-Se₂$, to 118.5 (2) and 103.6 (2) °, respectively. The local site symmetry for the manganese ions is very close to C_{2v} .

Within a linear chain the manganese ions are 4.471 A apart whereas the shortest Mn-Mn separation between manganese ions in adjacent chains in 6.164 **8,** the ions being separated by two selenium ions. Dominant magnetic interaction is expected within the linear chains, with only weak competing interchain interactions,

Equation 1 was applied in fitting the susceptibility data for $Ba₂MnSe₃$ and $Ba₂MnS₃$. The values of g and *J/k* were varied to obtain the best average fit of the theoretical curves to the susceptibility data for three separate preparations of each compound. The final values of g and J/k are given in Table VI and the re-

 $a_{\mu_{eff}} = 2.84x_m/T$. *b* θ from $x_m' = C/(T + \theta)$ —extrapolation of high-temperature $(>300^{\circ}K)$ $1/x_m$ ' *vs. T* data.

sulting theoretical curves are shown fitted to the experimental results in Figure 1. In view of the differences in susceptibilities for different sample preparations $(2-4\%)$ it was necessary to obtain an estimate of the errors associated with J/k and g. Values obtained from fitting eq 1 separately to the three sets of data differed up to 0.5° K in J/k and 0.03 in g and so these figures may be considered to be estimates of the errors associated with these quantities.

Also shown in Table VI are values of J/k and g obtained from application of the "reduced-spin" model, developed by Emori, *et a1.,12* in which the susceptibility is given by

$$
\chi_{\rm m}' = \frac{N g^2 \beta^2 S(S+1)}{3kT} \exp\left(\frac{J'}{kT}\right) \tag{2}
$$

and *J'* represents the exchange energy for a system composed of spins capable of orienting only parallel and antiparallel. The sign convention for antiferromagnetic exchange is negative in this theory. The true exchange integral, *J/k,* is given by

$$
\frac{J}{k} = \frac{\frac{3}{4}J'/k}{S(S+1)}
$$

Figure 3 shows a plot of log $(3k/N\beta^2)\chi_m'T$ *vs.* $1/T$ from

Figure 3.-Plot of susceptibility data using the "reduced-spin" model for (A) $Ba₂MnSe₃$ and (B) $Ba₂MnS₃$.

which g and J/k are obtained from the intercept $[-\log \{g^2 S(S + 1)\}]$ and the slope $[-J'/2.30k]$ of the straight line of best fit through the higher temperature $(>120^{\circ} K)$ results.

The fairly close agreement of the values of *J/k* and g with those obtained from the Heisenberg model indicates that eq *2* is a reasonable approximation for onedimensional antiferromagnetics.

For comparison, experimental g values are also given in Table 111. These were obtained from powder esr measurements on the compounds.

The values of the exchange integral obtained for $Ba₂MnS₃$ and $Ba₂MnSe₃$ are considerably greater than those obtained for CsMnCl₃.2H₂O $(J/k = -3.0^{\circ} K)$ and reflect the greater superexchange capacity of the chalcogenides relative to the halides. The exchange integral J/k is a measure of the overall exchange energy resulting both from direct exchange between the metal atoms and from superexchange *via* the bridging ligands. For the compounds $Ba₂MnX₃$, the large intrachain Mn-Mn separation $(\sim 4.5 \text{ Å})$ is expected to exclude direct exchange and the exchange energy may be considered to result predominantly from exchange *via* the bridging chalcogenide.

The strengths of the various superexchange mechanisms possible between the half-filled Mn^{2+} d orbitals and the anion p orbitals depend both on the polarizability of the anions and the geometry of the bridging system, *i.e.,* on the Mn-X separation and on the Mn-X-Mn angle. One cannot predict, *a priori,* the variation in the exchange in going from the selenium to the sulfur compound on the basis of covalency alone, especially when the angles remain identical in the two compounds. The complexity of determining the various contributions to the sign and magnitude of the net exchange has been discussed by Konamori^{14a} and by Huang and Orbach.^{14b} Thus in the chromium chalcogenide spinels^{14b} where antiferromagnetic interactions tend to increase from $O^{2-} \rightarrow S^{2-} \rightarrow S^{2-}$ the effect is offset by an increase in the ferromagnetic interactions along this series and the net effect is not predictable. In the series $ZnCr_2Se_4 \rightarrow ZnCr_2S_4 \rightarrow ZnCr_2O_4^{15}$ the antiferromagnetic Curie-Weiss constant, θ [°]K, becomes more negative so that the greater antiferromagnetic interaction in $Ba₂MnS₃$ is not unexpected although solely on the basis of the respective covalencies of the Mn-X bonds one would expect the superexchange to increase from the sulfide to the selenide. The value of Θ for $Ba₂MnS₃$ is also larger than for $Ba₂MnSe₃$.

We are studying the magnetic properties of the iron and cobalt compounds in order to gain further information concerning the exchange mechanisms present in these structures.

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